

## Study on the Pyrolysis Characteristic and the Microstructure of the Pyrolysis Products of $\beta$ Resins from Different Coal Tar Pitch

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**Summary:** Three types of  $\beta$  resins, namely,  $\beta$ -1,  $\beta$ -2, and  $\beta$ -3 were obtained from a medium coal tar pitch, modified pitch A, and modified pitch B, respectively. The pyrolysis characteristics of these three  $\beta$  resins were studied by TG/DTG analysis, FTIR spectroscopy, and curve-fitting method. The pyrolysis of  $\beta$ -1 showed a two-step mass loss, and the pyrolysis of  $\beta$ -2 showed a one-step mass loss. However, the pyrolysis of  $\beta$ -3 showed a three-step mass loss. This difference in pyrolysis among the three  $\beta$  resins was caused by the dissimilarity in their aromaticity index (Iar) and  $\text{CH}_3/\text{CH}_2$  ratio; these factors significantly affect the aromaticity and number of aliphatic side chains in  $\beta$  resin. The polarized microscopic structures of the pyrolysis products  $\beta$ -1-P,  $\beta$ -2-P, and  $\beta$ -3-P obtained from  $\beta$  resins show that  $\beta$ -1-P mainly has a coarse fibrous structure, whereas  $\beta$ -2-P and  $\beta$ -3-P have a mosaic structure and fine fibrous structure, respectively. The results of XRD and Raman analysis combined with curve-fitting method show that the content of ordered carbon microcrystalline structure in  $\beta$ -3-P is higher than the contents of other two samples, and  $\beta$ -2-P has the lowest content of this microcrystalline structure. In fact, the lower Iar ( $<0.5$ ) corresponded to a mosaic structure, and the higher Iar ( $>0.8$ ) corresponded to a fibrous structure. Moreover, oxidation linking reaction inhibited the formation of plane macromolecules, thus hindering the formation of an optically anisotropic fibrous structure during the pyrolysis of coal tar pitch.

Keywords: Coal tar pitch,  $\beta$  resins, Pyrolysis characteristic, Microscopic structure, Quantity analysis.

### Introduction

International Union of Pure and Applied Chemistry (IUPAC) define pitch as a solid material at room temperature, i.e., the residue obtained after the pyrolysis of organic materials or distillation of tar [1]. Coal tar pitch is a black, sticky solid material with a very high viscosity [2]. This also indicates that coal tar pitch is a complex material consisting of aromatic compounds with a broad molecular weight distribution [3]. Coal tar pitch is a promising candidate to produce carbon materials and carbon/carbon composites owing to its low cost and high performance to generate graphitizable carbons [4–6].

To obtain carbon materials or carbon/carbon composites with coal tar pitch as the matrix precursor, it is necessary to perform the deep processing of coal tar pitch [7, 8]. In previous studies [9–14], thermal treatment and air oxidation are the most common methods for the modification of coal tar pitch. These treatments cause the polymerization and condensation of pitch components, increasing their molecular weight [13]. Pitches show different compositions depending on the treatment [15]. The characterization of a modified pitch is fundamental to better understand their functional properties

according to the experimental conditions. Thermal analyses (TG, DTG, and DTA) are useful to characterize pitches as they easily distinguish pitches of different origins or allocated for different uses, or even distinguish pitches of the same origin, but prepared following different procedures [16, 17]. Fourier transforms infrared (FTIR) spectroscopy and curve-fitting methods are helpful to determine the pitch aromaticity and identify functional groups and aliphatic and aromatic hydrogens [18–24].

$\beta$  resin is a component of toluene insoluble and quinoline soluble (TI-QS) part of coal tar pitch.  $\beta$  resin has a narrow distribution of molecular weight, good fluidity, and good caking properties. All these properties endow  $\beta$  resin with a good reactivity and broad reaction temperature. In fact, the content of  $\beta$  resin is a prerequisite to determine the feasibility of coal tar pitch to produce high-quality carbon/graphite materials. The content of  $\beta$  resin plays a key role in the coal tar pitch processing phases [25, 26]. The property of  $\beta$  resin significantly affects the thermal treatment and air oxidation of coal tar pitch. The purpose of this study is to compare the pyrolysis behaviors and characteristics of  $\beta$  resins obtained from a medium coal tar pitch and modified pitches

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obtained by thermal treatment and air oxidation by TG/DTG analysis, FTIR spectroscopy, and curve-fitting method. In addition, the microstructural characteristics of pyrolysis products obtained from  $\beta$  resins were successfully evaluated by various analysis methods.

## Experimental

### Materials

The medium coal tar pitch was purchased from Shanxi Coking Plant (Shanxi, China). Toluene and quinoline were purchased from Tianjin Regent Chemicals (Tianjin, China). The proximate analysis of medium coal tar pitch is shown in Table-1.

### Preparation of modified pitch

A refined pitch prepared from the medium pitch was used as the raw material for preparing the modified pitches. Modified pitch A was obtained by air oxidation in a stainless steel reactor. The refined pitch was heated to 280 °C at a heating rate of 5 °C/min, the holding time was 6 h, and air was used as the oxygen supply agent. Thermal treatment was used to prepare modified pitch B. Briefly, 100 g of refined pitch was heated to 420 °C at a heating rate of 5 °C/min, the holding time was 4 h, and N<sub>2</sub> was used as the protection gas. The proximate analysis of refined pitch and modified pitches is shown in Table-1.

Table-1: Proximate Analysis of pitch Samples.

Sample	SP <sup>a</sup> /°C	TI <sup>b</sup> /wt%	QI <sup>c</sup> /wt%	CV <sup>d</sup> /wt%	$\beta$ resin <sup>e</sup> /wt%
Medium pitch	73	19.65	5.32	49.15	14.33
Refined pitch	49	10.95	0.07	41.76	10.88
Modified pitch A	232	53.16	30.18	52.33	22.98
Modified pitch B	228	60.29	25.39	81.88	34.90

<sup>a</sup> Softening point, <sup>b</sup> Toluene insoluble, <sup>c</sup> Quinoline insoluble, <sup>d</sup> Coke value, <sup>e</sup> TI-QS

### Preparation of $\beta$ resins

Three types of  $\beta$  resins, namely,  $\beta$ -1,  $\beta$ -2 and  $\beta$ -3 were obtained from the medium coal tar pitch, modified pitch A and modified pitch B. The  $\beta$  resin was obtained using a two-step solvent extraction method at 90 °C. In the first step, the toluene-insoluble part (TI) in the coal tar pitch was obtained using toluene as the solvent. Briefly, 30 g of coal tar pitch and 600 mL of toluene as the solvent were charged into a three-neck flask, and the coal tar pitch was extracted for 3 h under stirring at 90 °C. The mixture was filtered, and the residue was TI. In

the second step, the quinoline-soluble part (QS) in the TI was obtained using quinoline as the solvent. The extraction and filtration processes were the same as above. The filtrate was distilled at 240 °C. The distillation residue was  $\beta$  resin.

### Characterization of $\beta$ resins

#### Ultimate analysis of $\beta$ resins

The ultimate analysis of  $\beta$  resins was performed using a Vario EL  $\beta$  elemental analyzer (Elementar, Germany). Oxygen content was calculated using the difference method. Table-2 shows the ultimate analysis data of  $\beta$  resins.

Table-2: Ultimate analysis of  $\beta$  resins.

Samples	C/%	H/%	N/%	S/%	O/%	C/H
$\beta$ -1	92.22	3.78	1.25	0.48	2.27	2.03
$\beta$ -2	91.95	3.81	1.22	0.38	2.64	2.01
$\beta$ -3	93.05	3.93	1.20	0.37	1.45	1.97

#### Thermogravimetric analysis (TGA)

TGA and derivative thermogravimetry (DTG) measurements (TAQ500, USA) were carried out under a N<sub>2</sub> flow rate of 100 mL min<sup>-1</sup>, the samples mass was 10 mg, the temperature ranged from 20 °C to 650 °C, and the heating rate was 10 °C min<sup>-1</sup>. Fig 3 shows the TG/DTG analysis of  $\beta$  resins.

#### FTIR spectroscopy

The FTIR spectra of  $\beta$  resins were obtained using a NICOLETIS5 FT-IR spectrometer at a resolution of 4 cm<sup>-1</sup>. Prior to the FTIR analysis, KBr pellets were prepared by grinding a mixture of 1 mg of sample with 150 mg of KBr. Duplicate pellets were used for the quantitative measurements. The aliphatic (Hal) contents were calculated from the integrated absorbance areas of the bands at 3000–2700 cm<sup>-1</sup>. The bands in the IR spectra were assigned according to the literature [27, 28].

#### Characterization of pyrolysis products

##### Polarizing microscope analysis

A polarizing microscope (Axio Scope A1 pol, Carl Zeiss, Germany) was used to characterize the optical microstructure of samples. Briefly, the pyrolysis products of  $\beta$  resin ( $\beta$ -1-P,  $\beta$ -2-P, and  $\beta$ -3-P) obtained at 650 °C was embedded in epoxy resin, and the micro-object was obtained by sanding and polishing. Then, the sample was placed under a microscope with

two drops of cedar oil. The graphs of samples obtained from the polarizing microscope are shown in Fig. 4.

#### XRD analysis

$\beta$ -1-P,  $\beta$ -2-P, and  $\beta$ -3-P were characterized by powder X-ray diffraction (XRD) analysis with a tube voltage of 40 kV, tube current of 40 mA, scanning range of 10–90°, and step size of 0.013°.

#### Raman spectroscopic analysis

The carbon microcrystalline structures of  $\beta$ -1-P,  $\beta$ -2-P, and  $\beta$ -3-P were determined by Raman spectroscopy (LabRAM HR Evolution, JOBIN YVON, France) with an incident wavelength of 532 nm. The distribution and contents of carbon microcrystalline structures were calculated from the integrated intensity areas of the bands at 900–1800  $\text{cm}^{-1}$ . The bands in the Raman spectra were assigned according to the literature [29–32], as shown in Table-4. Briefly, D3 band was fitted by a Gaussian function, but the other four bands were fitted by a Lorentz function.

#### Curve-fitting analysis

Selected zones of the FTIR and Raman spectra were studied by curve-fitting analysis using a commercially available data-processing program (Origin 7.0). Prior to the curve-fitting analysis, background corrections were applied as described in the literature [27, 28, 33]. The initial approximation of the number of bands and peak positions was obtained by determining the second derivatives of the spectral data, and Gaussian and Lorentz functions were used as the mathematical functions [34].

To determine the goodness of fit criteria, the following aspects were considered: coefficient of determination (COD) and standard errors of parameters (chi-squared). Briefly, all the COD values of the curve-fitting analysis in all the cases were above 0.999, and the chi-squared was  $<10^{-5}$ .

## Results and discussion

#### Qualitative and quantitative FTIR analysis of $\beta$ resins

Table-3: Raman bands and vibration modes reported by literatures [29-32].

Band	Raman shift ( $\text{cm}^{-1}$ )	Vibration mode
G	1580	Ideal graphitic lattice ( $E_{2g}$ -symmetry)
D1	1350	Disordered graphitic lattice (graphene layer edges, $A_{1g}$ symmetry)
D2	1620	Disordered graphitic lattice (surface graphene layers, $E_{2g}$ -symmetry)
D3	1500	Amorphous carbon (Gaussian line shape)
D4	1200	Disordered graphitic lattice ( $A_{1g}$ symmetry), polyenes, ionic impurities

FTIR Spectroscopy is one of the most common techniques to achieve information about functional groups in complex solids. This has also been used as an effective tool for the quantitative estimation of structures of pitches combined with the curve-fitting method. Fig. 1 shows the FTIR spectra of  $\beta$  resins obtained from different coal tar pitches. The peak at 3050  $\text{cm}^{-1}$  can be attributed to the aromatic stretching mode, and the peaks between 3000–2700  $\text{cm}^{-1}$  can be attributed to the aliphatic C–H stretching mode. Peaks in the region 900–700  $\text{cm}^{-1}$  can be attributed to the aromatic C–H out-of-plane bending mode.

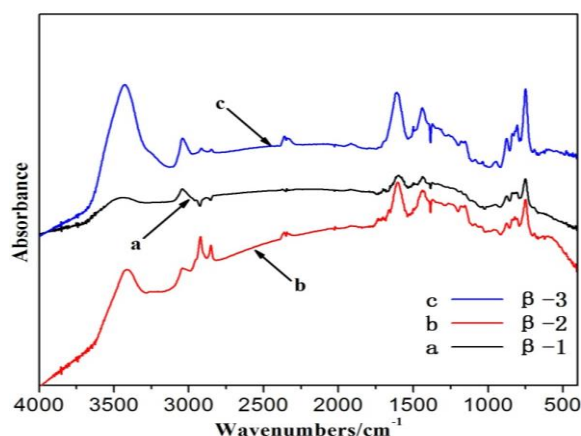


Fig. 1: FTIR spectra of  $\beta$  resins.

To further evaluate the impurity of  $\beta$  resins from different coal tar pitches, the 3000–2800  $\text{cm}^{-1}$  was studied by curve-fitting analysis. A series of relationships defined as the ratios of integrated absorbance areas were used to quantify the structural changes from the curve-fitting bands. Iar is the aromatic hydrogen index, and  $\text{CH}_3/\text{CH}_2$  ratio is the aliphatic side chain index, obtained from the ratios of integrated absorbance peak areas. For example, Abs :3050 is the integration area of absorbance in 3050  $\text{cm}^{-1}$  fitting peak. The relationships can be expressed as follows:

$$I_{ar} = \text{Abs} \cdot 3050 / (\text{Abs} \cdot 3050 + \text{Abs} \cdot 2920) \quad (1)$$

$$\text{CH}_3/\text{CH}_2 = 2950 \text{ cm}^{-1} \text{ band} / 2920 \text{ cm}^{-1} \text{ band} \quad (2)$$

Equation 1 shows the aromaticity of  $\beta$  resin. Equation 2 shows the ratio of methyl/methylene structures, and it can be considered as an estimate of the length of aliphatic chains of  $\beta$  resin [22, 34].

Table-3 shows the aromaticity index of  $\beta$  resins. A clear difference was observed among the three  $\beta$  resins. The aromaticity index of  $\beta$ -1 and  $\beta$ -3 were both  $>0.85$ , indicating that these two  $\beta$  resins have a high aromaticity. However, the aromaticity index of  $\beta$ -2 was  $<0.5$ . This indicates that  $\beta$ -2 has a lower aromaticity, but a higher content of aliphatic hydrogen. The data shown in Table-4 also indicate that the integration area of aliphatic C-H stretching mode in  $\beta$ -2 resin was five times higher than those of the other two  $\beta$  resins.

Table-4: The index of  $I_{ar}$  on  $\beta$  resins.

Samples	Centre/cm <sup>-1</sup>	Height <sup>1</sup>	Width/cm <sup>-1</sup>	Area <sup>2</sup>	$I_{ar}$
$\beta$ -1	3043	0.01616	64.604	1.12021	0.85
	2925	0.00908	20.249	0.19756	
$\beta$ -2	3044	0.01362	48.212	0.66933	0.46
	2921	0.03028	22.178	0.78575	
$\beta$ -3	3043	0.03033	53.033	1.64343	0.91
	2918	0.00584	24.106	0.16129	

<sup>1</sup>Absorbance, <sup>2</sup>Integrate absorbance unit  $\times \text{cm}^{-1}$

To better understand the distribution of aliphatic hydrogen and number of alkyl chains in  $\beta$  resins, the curve-fitting analysis of 3000–2700  $\text{cm}^{-1}$  was carried out. The aliphatic region of raw coal tar pitch can be curve-fitted to a series of five bands attributed to asymmetric  $-\text{CH}_3$  and  $-\text{CH}_2-$  stretching (2953 and 2923  $\text{cm}^{-1}$ , respectively), symmetric  $-\text{CH}_3$  and  $-\text{CH}_2-$  stretching (2879 and 2853  $\text{cm}^{-1}$ , respectively), and methine C-H stretching (2896  $\text{cm}^{-1}$ ), as shown in Fig. 2. The results are shown in Table-5.

Table-5: Curve-Fitting for the Aliphatic C-H Stretching Bands of  $\beta$  Resin from Different Pitches.

Centre(cm <sup>-1</sup> )	Assignment	Width(cm <sup>-1</sup> )			Height <sup>1</sup>			Area <sup>2</sup>		
		$\beta$ -1	$\beta$ -2	$\beta$ -3	$\beta$ -1	$\beta$ -2	$\beta$ -3	$\beta$ -1	$\beta$ -2	$\beta$ -3
2853	Sym. $\text{R}_2\text{CH}_2$	27.524	25.596	28.919	0.005	0.015	0.003	0.087	0.220	0.053
2870	Sym. $\text{RCH}_3$	33.091	67.454	83.767	0.002	0.009	0.002	0.039	0.338	0.102
2896	$\text{R}_3\text{CH}$	79.138	55.759	59.571	0.003	0.013	0.003	0.138	0.433	0.100
2923	Asym. $\text{R}_2\text{CH}_2$	39.147	39.502	41.519	0.011	0.034	0.006	0.250	0.780	0.151
2953	Asym. $\text{RCH}_3$	52.875	57.035	41.519	0.004	0.012	0.001	0.138	0.398	0.028

<sup>1</sup>Absorbanc, <sup>2</sup>Integrate absorbance unit  $\times \text{cm}^{-1}$

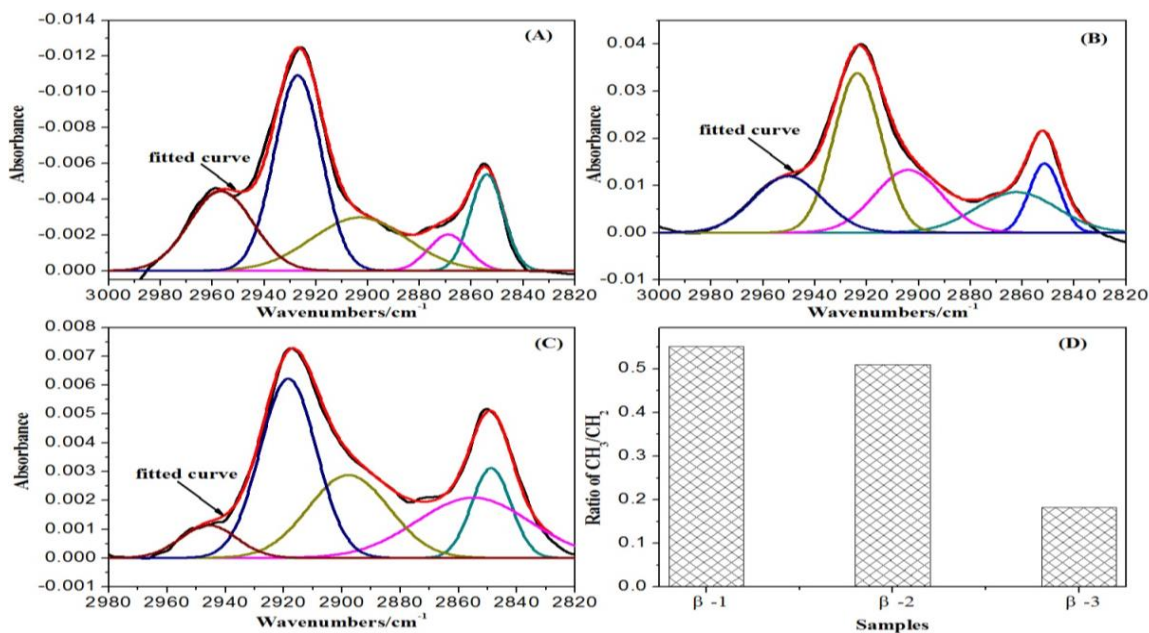


Fig. 2: Curve-fitted IR spectrum of the aliphatic C-H stretching bands for  $\beta$  resin from different pitches: (A)  $\beta$ -1, (B)  $\beta$ -2, (C)  $\beta$ -3, (D) The ratio of  $\text{CH}_3/\text{CH}_2$  of samples.

According to the literature [34], the  $\text{CH}_3/\text{CH}_2$  ratio reflects the length of aliphatic chain and branched chain. The changes in the aliphatic bands of three  $\beta$  resins ( $\text{CH}_3/\text{CH}_2$  ratio shown in Fig. 2 (D)) indicate that the ratio of  $\beta$ -3 resin is significantly lower than other two  $\beta$  resins. This shows that the length of aliphatic chain of  $\beta$ -3 is longer, and the length of aliphatic branched chains is also longer.

Compared to  $\beta$ -1 and  $\beta$ -2,  $\beta$ -3 has the highest aromaticity and most aliphatic chain [Table-4 and Fig. 2 (D)]. This is probably caused by the high condensation degree in  $\beta$ -3. Air oxidation destroyed the aromatic C=C bonds, while hindering the growth of aliphatic chains [Fig. 2 (D)]. Therefore,  $\beta$ -2 extracted from modified pitch A has the lowest Iar (Table-4).

### TGA

Thermal analysis methods are the most promising methods to study the pyrolysis behavior of pitches and characterize them. Thermal analysis methods are efficient to evaluate the nature of the reaction. TGA provides information about the mass loss associated with reactions. TGA and DTG also provide information about mass loss and the rate of mass loss at a particular temperature.

Initially, the pyrolysis of  $\beta$ -1,  $\beta$ -2, and  $\beta$ -3 was carried out by TGA to study the different pyrolysis characteristics of  $\beta$  resins obtained from different pitches. The TGA and DTG curves (Fig. 3) show a significant difference in the pyrolysis characteristics of  $\beta$  resins.

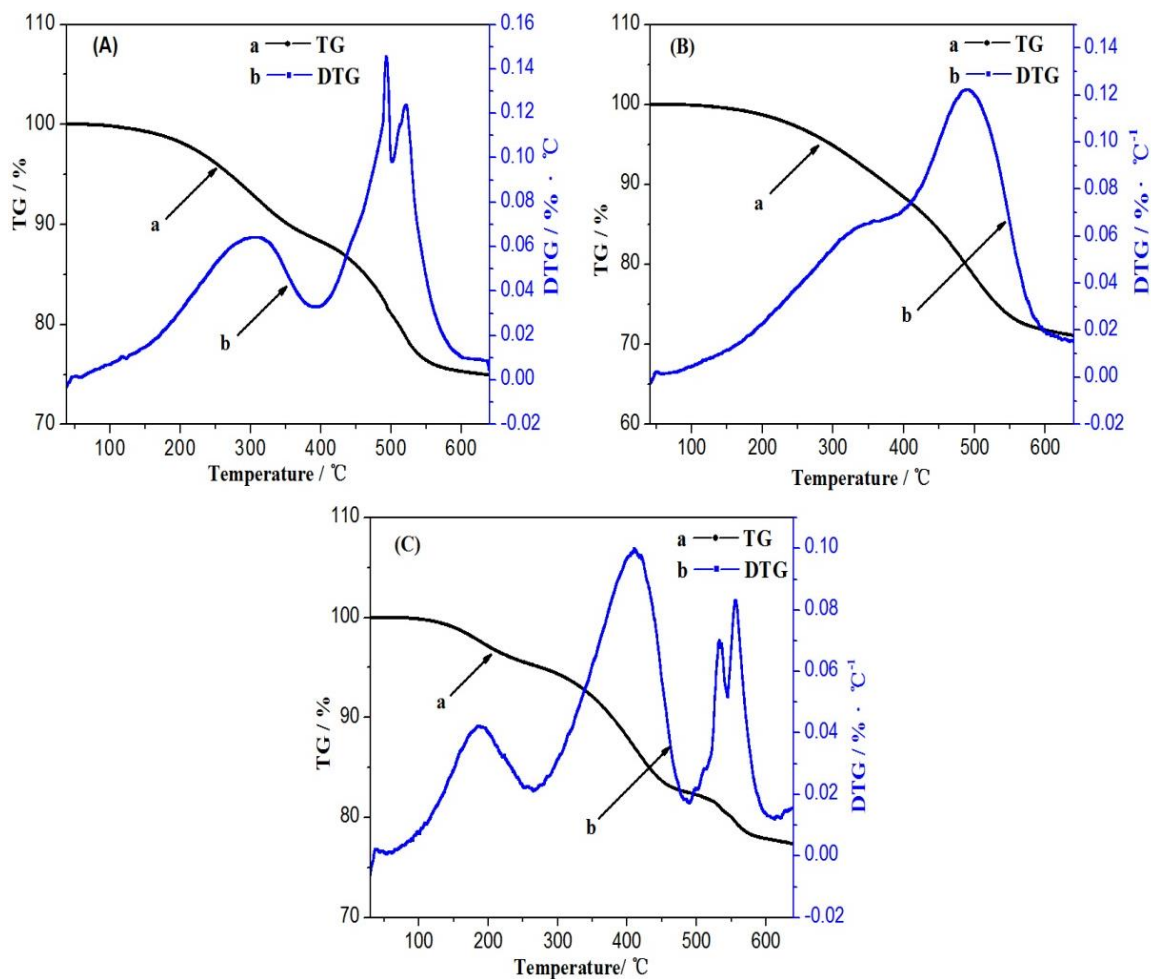


Fig. 3: TG/DTG analysis of  $\beta$  resins: (A)  $\beta$ -1, (B)  $\beta$ -2, (C)  $\beta$ -3.

The TG and DTG curves (Fig. 3 A) show that  $\beta$ -1 mass loss clearly occurs in two separate steps. In the first step (180–400 °C),  $\beta$ -1 loses 11.78 wt% of mass, and the temperature of maximum mass loss ( $T_{\max}$ ) was 285 °C. In the second step (450–600 °C), the  $T_{\max}$  is 493 °C. This indicates that  $\beta$ -1 contains two different components of different thermal stabilities and compositions. In fact, the elemental analysis shows that the total carbon, oxygen, and hydrogen contents account for 98.27 wt% of the total content (Table-2), the other components being sulfur and nitrogen. On the other hand, the FTIR spectra (Figs. 1 and 2), Iar index (Table-4), and  $\text{CH}_3/\text{CH}_2$  ratio (Fig. 2D) show that  $\beta$ -1 contains mainly an aromatic ring structure and secondly aliphatic side chains. In other words, the molecular structure of  $\beta$ -1 was composed of a large number of polycyclic aromatic molecules with some aliphatic side chains. When the temperature was increased to 180–400 °C, the aliphatic side chains began to fracture, and the first step of mass loss occurred with the escape of small molecules of poor thermal stability. Molecules with a higher thermal stability escaped as the temperature was increased to 450–600 °C.

Fig. 3 B shows that  $\beta$ -2 mass loss is completed in a single step in a wide temperature range 200–600 °C. The carbonaceous residue is 68.75 wt% at 800 °C, and the  $T_{\max}$  is 493 °C. The Iar index (Table-4), curve-fitted IR spectrum and data at the zone 3000–2800  $\text{cm}^{-1}$  (Fig. 2 B and Table-5),  $\text{CH}_3/\text{CH}_2$  ratio (Fig. 2D) indicate that  $\beta$ -2 was composed of mainly aliphatic side chains and secondly aromatic ring structures. This is because of the characteristics of modified pitch A, which was obtained by air oxidation. In the presence of oxygen, the aromatic structure of coal tar pitch was destroyed; therefore, the CV content also decreased (Table-1). This also narrowed the molecular distribution. Because of these structural features, the pyrolysis of  $\beta$ -2 showed a single weight-loss peak with a wide weight-loss temperature range.

$\beta$ -3 weight loss was completed in three separate steps. In the first step (150–250 °C),  $\beta$ -3 lost 5 wt% of the total weight, and the  $T_{\max}$  was 220 °C. In the second step (260–480 °C), the weight loss was completed, and the  $T_{\max}$  was 419 °C. In the third step (490–600 °C), the weight loss occurred in a narrow temperature range, and the  $T_{\max}$  was 556 °C. Compared to  $\beta$ -1 and  $\beta$ -2, the ratio of C/H on  $\beta$ -3 clearly decreased (Table-2). Additional information was obtained by calculation using the FTIR spectra

(Fig. 1) combined with curve-fitting at the zone of aliphatic C–H stretching bands for  $\beta$ -3 (Fig. 2 C). The curve-fitting data are shown in Table-5. The Iar index (Table-4) and  $\text{CH}_3/\text{CH}_2$  ratio (Fig. 2D) show that  $\beta$ -3 has a structure with a high aromaticity index and more aliphatic side chains. These results indicate a cross-linked macromolecular structure in  $\beta$ -3 that occurred in the modification of modified pitch B. During the pyrolysis, the cross-linked bonds were destroyed at a lower temperature. This resulted in residues made up of two components of different thermal stabilities. With persistent overheating, the pyrolysis of two components occurs at different temperature ranges.

#### *Polarized microscopic structure analysis of pyrolysis products*

The polarized microscopic images of pyrolysis products obtained from  $\beta$  resins showed significant differences (Fig. 4). Figs. 4 (A) and (B) clearly show that  $\beta$ -1-P had mainly a coarse fiber structure. However, the polarized microscopic images of  $\beta$ -2-P and  $\beta$ -3-P showed a mosaic structure and fine fibrous structure, respectively. These are probably related to the aromaticity index (Table-3). Briefly, the lower Iar (<0.5) corresponds to a mosaic structure, and the higher Iar (>0.8) corresponds to a fibrous structure. Another reason for this difference is the source of  $\beta$  resins. In fact,  $\beta$ -2 was obtained from modified pitch A. However, modified pitch A was prepared by air oxidation, and the O<sub>2</sub> present in the system caused the oxidation linking reaction. The oxidation linking reaction inhibits the formation of planar macromolecules in modified pitch A. Therefore, the pyrolysis of  $\beta$ -2 could not create an optical anisotropic fibrous structure, but a mosaic structure.

#### *XRD analysis of samples*

The microstructures of samples were determined by XRD analyses. The XRD patterns of the samples are shown in Fig. 5 (A). The trends of the XRD patterns of the three samples are similar. However,  $\beta$ -3 has the largest intensity of (002) and the narrowest width of (002). Compared to  $\beta$ -1-P,  $\beta$ -2-P has the maximum width of (002). In fact, compared to  $\beta$ -1-P and  $\beta$ -2-P,  $\beta$ -3-P has the most ordered microcrystalline structure.

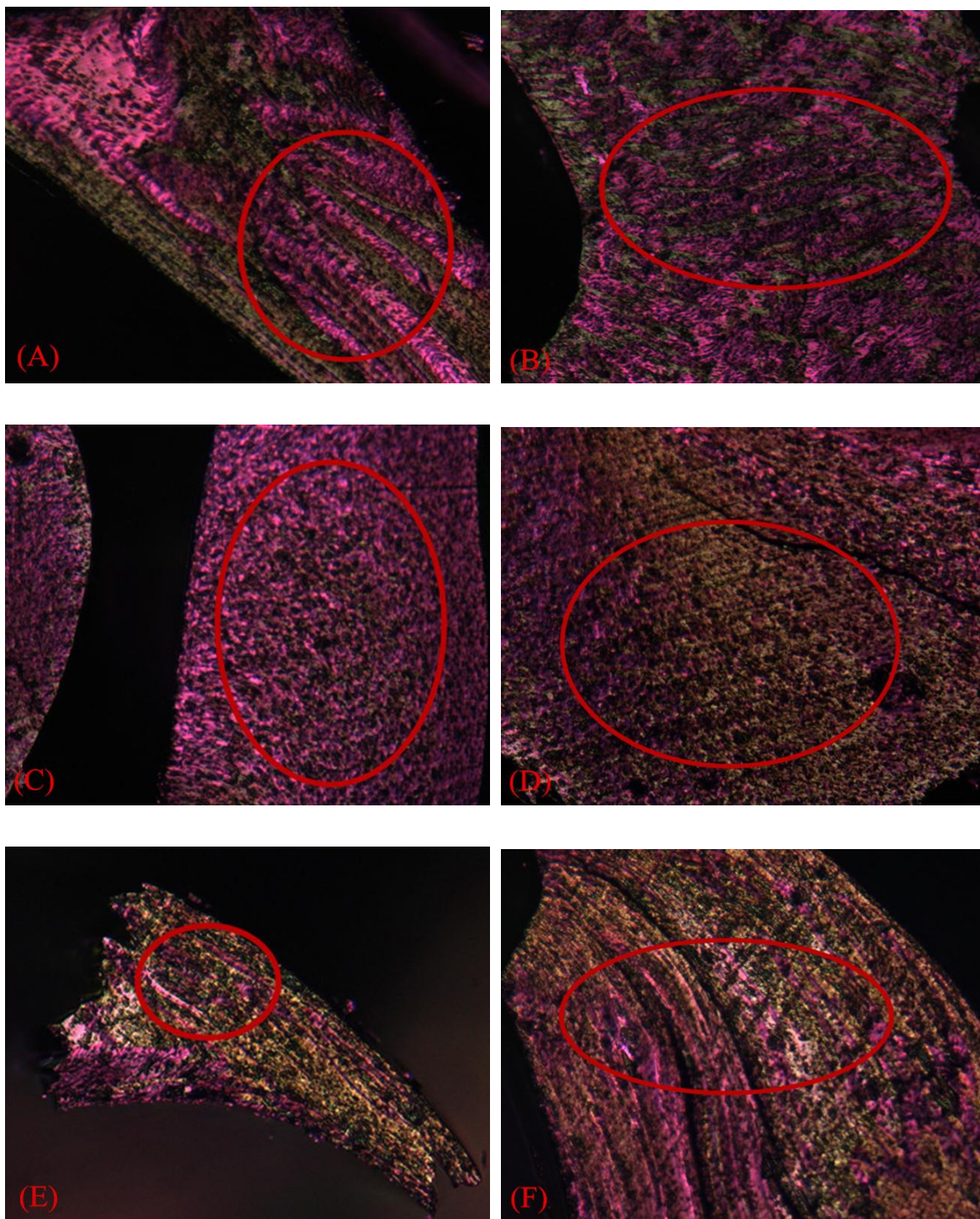


Fig. 4: Polarized microscopic structure analysis of pyrolysis products from  $\beta$  resin, (A) and (B) for  $\beta$ -1-P, (C) and (D) for  $\beta$ -2-P, (E) and (F) for  $\beta$ -3-P.

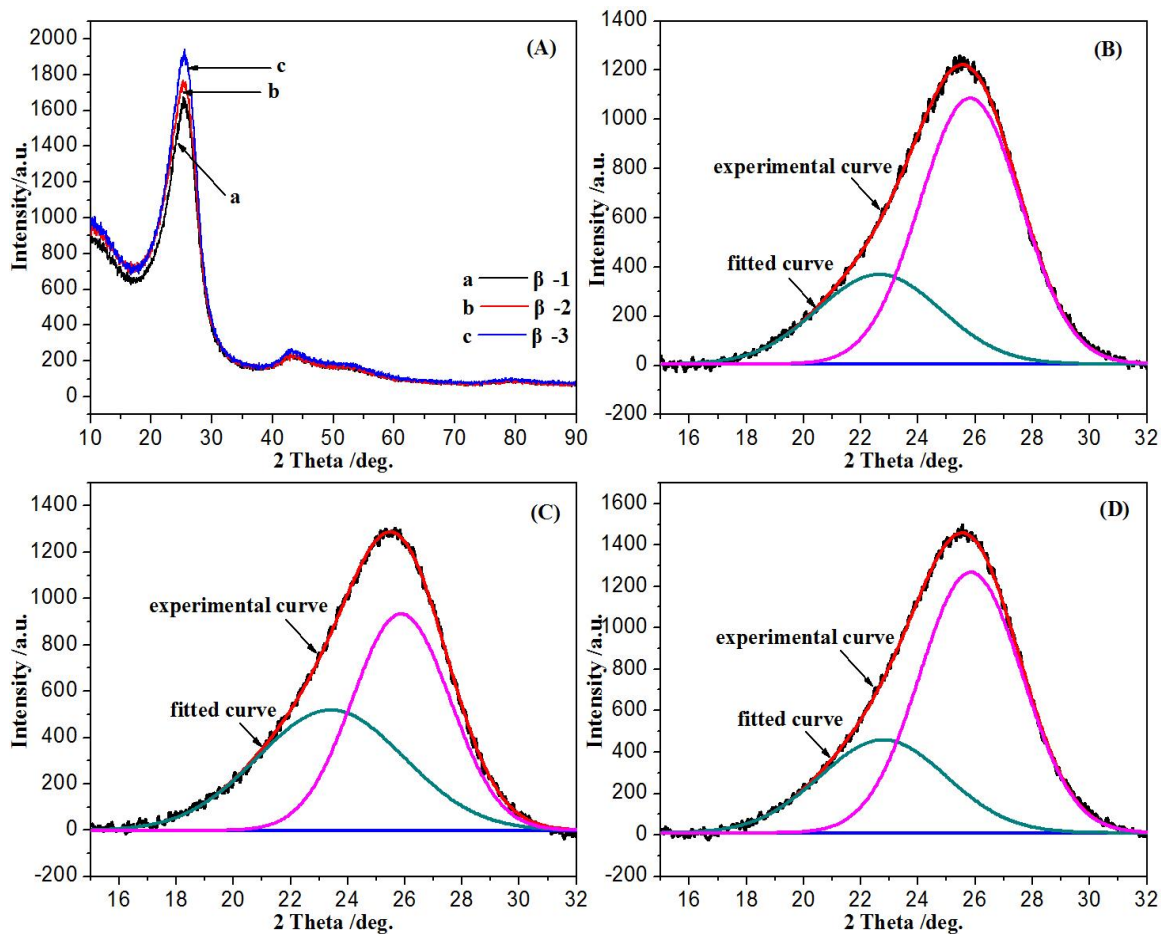


Fig. 5: XRD analyses of samples (A), and curve-fitted graph of pyrolysis product from  $\beta$  resins: (B) for  $\beta$ -1-P, (C) for  $\beta$ -2-P and (D) for  $\beta$ -3-P.

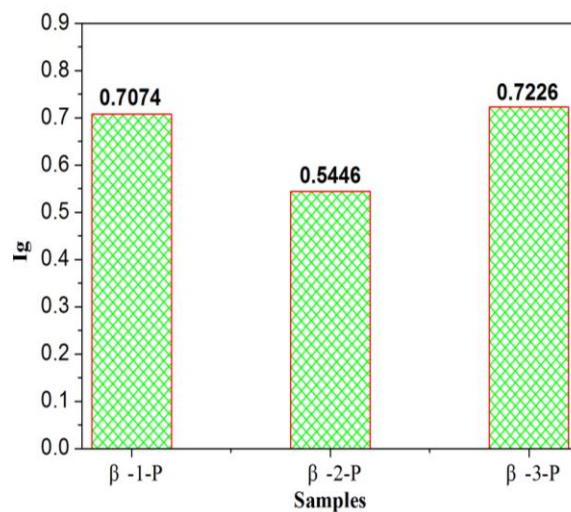


Fig. 6: The distribution of  $I_g$  on three samples.

For a better comparison of the microcrystalline structure in the samples, the curve-fitting method has been used. Researchers [35] found that the broad hump of (002) of carbon material can be divided into two Gaussian peaks at  $20^\circ$  and  $26^\circ$  nearby peak-fitting, namely,  $\gamma$ -band and  $\pi$ -band, respectively. The areas of the  $\gamma$  and  $\pi$  peaks are probably equal to the content of amorphous carbon and trended ideal graphite carbon. The content of trended ideal graphite carbon ( $I_g$ ) can be calculated using the following equation:

$$I_g = A(\pi)/[A(\pi)+A(\gamma)] \quad (3)$$

$A(\pi)$  and  $A(\gamma)$  are the area of  $\pi$  and  $\gamma$  peaks after the peak-fitting.

The curve-fitting profile of  $\beta$  resins is shown in Figs 5(B), (C), and (D); the distribution of  $I_g$  on the samples is shown in Fig. 6. Obviously, the numerical of  $I_g$  in  $\beta$ -3-P is higher than  $\beta$ -1-P, and much higher than that in  $\beta$ -2-P. This indicates that the contents of trended ideal graphite carbon of  $\beta$ -3-P and  $\beta$ -1-P were much higher than  $\beta$ -2-P at the same pyrolysis temperature. In other words, the graphitization of  $\beta$ -3-P should be easier than  $\beta$ -2-P. This phenomenon can be explained by the  $I_{ar}$  of the samples. A higher aromaticity indicates more planar

macromolecules in the molecular components of the sample, easy to graphitize.

#### Raman spectroscopic analysis of samples

As shown in Fig. 7(A), the two typical bands, the G band (graphite band) and D band (disorder band) were detected in the first-order region in all the studied samples. The Raman spectra obtained from  $\beta$ -1-P and  $\beta$ -3-P showed a higher intensity than  $\beta$ -2-P. To determine the carbon crystalline structure of the samples, the curve-fitting method was used to obtain more information from the Raman spectrum [Figs. 7 (B), (C), and (D)]. The fitting standards are shown in Table-3.

The band area ratios ( $I_G/I_{All}$ ,  $I_{D3}/I_G$ ) were used to measure the content of microcrystalline structure in samples (Table-6).  $\beta$ -3-P has the maximum  $I_G/I_{All}$  ratio and minimum  $I_{D3}/I_G$  ratio. However, these two parameters of  $\beta$ -2-P are contrary to  $\beta$ -3-P. This indicates that the content of well-ordered carbon structure in  $\beta$ -3-P was higher than the other two samples. In contrast,  $\beta$ -2-P has the highest level of amorphous carbon. This phenomenon is identical with the optical microstructure of three samples.

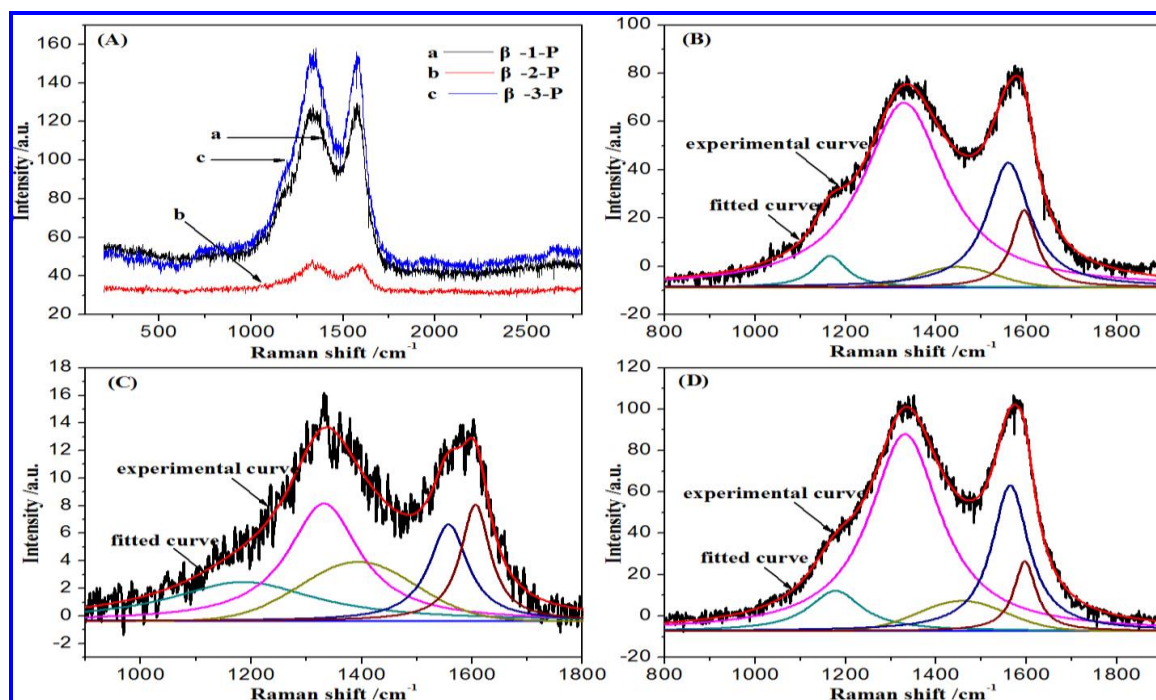


Fig. 7: Raman spectrum analyses of samples (A), and curve-fitted graph of pyrolysis product from  $\beta$  resins: (B) for  $\beta$ -1-P, (C) for  $\beta$ -2-P and (D) for  $\beta$ -3-P.

Table-6: Curve-fitting data of the first-order region for samples.

Samples	Integrate Area				Ratio		
	I <sub>D1</sub>	I <sub>D2</sub>	I <sub>D3</sub>	I <sub>D4</sub>	I <sub>G</sub>	I <sub>G</sub> /I <sub>M</sub>	I <sub>D3</sub> /I <sub>G</sub>
β-1-P	15483.98	2121.44	1202.73	1113.55	5721.16	0.22	0.21
β-2-P	1280.43	552.96	746.98	766.19	587.76	0.15	1.27
β-3-P	16533.92	1905.86	1360.39	2644.46	6838.28	0.23	0.19

### Conclusions

The pyrolysis characteristics of different β resins obtained from different coal tar pitches were evaluated experimentally. Modification methods play a key role on the molecular structure of β resins. FTIR spectroscopic analysis combined with the curve-fitting method showed that the I<sub>ar</sub> and CH<sub>3</sub>/CH<sub>2</sub> ratio of β resins have significant differences. This indicates that β-1 was mainly composed of an aromatic ring structure and secondly aliphatic side chains. β-2 was mainly composed of aliphatic side chains and secondly aromatic ring structures, and β-3 has a high aromaticity index and more aliphatic side chains. These compositions of molecular structure contribute to the two clear weight-loss steps for β-1, a single weight-loss step for β-2, but three weight-loss steps for β-3 during the pyrolysis. Moreover, the microstructures of pyrolysis products obtained from β resins varied. The polarized microscopic images of β-1-P, β-2-P, and β-3-P showed a coarse fibrous structure, mosaic structure, and fine fibrous structure, respectively. The XRD and Raman spectroscopic analyses indicate that the content of well-ordered carbon structure in β-3-P was higher than those of the other two samples, and β-2-P has the highest level of amorphous carbon. This indicates that β-3-P has a good graphitizability than the other two samples. This further proves that thermal treatment is a perfect method to produce a coal tar pitch-based carbon precursor with good graphitizability, and air oxidation is an efficient method to produce an isotropic coal tar pitch-based carbon precursor.

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